

Preparation and characterization of graphene-based fluorine doped tin dioxide thin films via spray pyrolysis technique

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In this work, fluorine-doped tin oxide (FTO) and graphene/fluorine-doped (G-FTO) thin films were prepared using a low-cost spray pyrolysis method at a substrate temperature of 500 °C. For the FTOs, stannous chloride was dissolved in methanol and acetic acid to form the precursor solution. A 0.05 mole (M) of hydrofluoric acid was added to the precursor as an n-type impurity. The FTO thin film has an optical transmittance of 82% and electrical sheet resistance of 15 Ω/□. By meticulously integrating graphene into the optimal precursor solution of FTO, a significant improvement in the electrical conductivity of the prepared samples was achieved, leading to a reduction in the sheet resistance to 8 Ω/□ with a suitable optical transmittance of 79%. Structural, morphological, optical, and electrical properties of the prepared sample are investigated using X-ray diffraction, scanning electron microscope, UV spectroscopy, and four-point probe technique. The best performance of the FTO thin films is achieved utilizing 2.5 μmole/L of fluorine concentration at a substrate temperature of 500°C for a spraying exposure time of 20 min. The prepared sample has an electrical sheet resistance of 15 Ω/□, optical transmittance of 82%, and figure-of-merit of $91.2 \times 10^{-4} \Omega^{-1}$. The addition of 0.4 μmole/L of graphene to the optimum FTO samples enhances the performance by a remarkable reduction in the electrical sheet resistance to 8 Ω/□ and an acceptable reduction in the optical transmittance of 79%. The overall value of the figure-of-merit increased to $118.3 \times 10^{-4} \Omega^{-1}$. The achieved results offer a high potential for adopting the prepared films for electronic and optoelectronic applications.

Keywords: fluorine-doped tin-oxide, graphene/fluorine-doped tin-oxide, nebulizer spray pyrolysis technique, sheet resistance, transmittance

1 Introduction

Recently, transparent conducting oxide (TCO) thin films inhabit an enormous area in the field of optoelectronic devices owing to their distinctive properties such as high electrical conductivity and visible light transparency [1-5]. SnO₂ has been identified as a particularly suitable transparent conducting oxide material due to its chemical stability, hard mechanical properties, and ability to withstand elevated temperatures. [6]. An effective method to further improve the electrical and optical properties of SnO₂ is doping with foreign impurities such as fluorine, antimony, thorium, indium, etc [7]. Fluorine-doped tin-oxide (FTO) has been recently used in many applications such as protective and anti-reflection coating, gas sensors, photovoltaic cells [8], and liquid crystal display (LCD) as transparent electrodes [9]. SnO₂:F thin films can be synthesized by various techniques. The most commonly used techniques include chemical vapor deposition [10], radio-frequency magnetron sputtering [11], sol-gel [12], pulsed laser deposition [13], and spray pyrolysis deposition technique [14]. Seeking the most reliable and economical deposition

technique is a considerable goal. Among these fabrication methods, spray pyrolysis is a convenient, low-cost, and powerful tool to produce various kinds of thin films. When compared to other methods of deposition, spray pyrolysis has several benefits, such as its simplicity, affordability, reproducibility, high growth rate, applicability to mass production, and the capability to uniformly coat large areas for industrial optoelectronic applications [3].

In this work, fluorine-doped tin oxide and graphene-fluorine-doped tin oxide (G-FTO) thin films were fabricated using the spray technique to improve both electrical and optical properties. The graphene addition to the main precursor of FTO enhances the overall performance of the prepared thin films. The high electrical conductivity and optical transparency of the prepared sample of FTO and G-FTO push it to be a good candidate in solar cell applications. The transparent conducting oxide (TCO) layer such as FTO, ITO and ZnO is a very important layer in solar cells. It must have a high electrical conductivity to collect the generated current and high optical transmission to pass a larger amount of the

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incident sun light. In this work, the influence of fluorine and graphene concentrations on the morphological structure, electrical, and optical properties of the produced thin films is being studied in detail. High-quality FTO and G-FTO thin film coats prepared at optimal preparation conditions could be used in optoelectronic applications efficiently.

2 Experimental work

2.1 Preparation of spray solution

Fluorine-doped tin-oxide thin films grown on glass substrates are prepared using a nebulizer spray technique at a temperature of 500°C. The initial solution is prepared by dissolving 2 g of stannous chloride ($\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$) in methanol as a solvent and glacial acetic acid (Sigma Aldrich) as an ionizing agent. The fluorine doping as an n-type dopant is achieved by adding 0.05M of hydrofluoric acid to the preliminary solution. The combination is agitated at 70°C for one hour in a sealed vessel until the powders have been fully dissolved. The glass substrates are cleaned by utilizing a combination of deionized water and methanol with the assistance of ultrasonication for 30 minutes.

Graphene is composed of a single layer of carbon atoms arranged in a hexagonal lattice. The material exhibits incredibly strong and lightweight properties. Moreover, it shows excellent electrical and thermal conductivity making it a promising candidate for a wide range of applications, including electronics, energy storage, and biomedical devices. Graphene is often referred to as the “basic building block” of many other carbon-based materials, including carbon nanotubes, fullerenes, and graphite. Its unique properties make it an exciting area of research, with many potential applications yet to be discovered [15-16]. In recent years, there has been a fast-growing interest in the use of graphene due to its electrical, optical, mechanical, and thermal properties [17-18]. A wide range of applications are using graphene specifically in the 6G wireless communication systems and the terahertz frequency band [19-20]. Practically, graphene could be synthesized in various ways on different substrates. This study involves the preparation of a highly concentrated surfactant dispersion of graphene through ultrasonication and centrifugation. The critical parameters affecting the concentration and dispersion quality are graphite concentration, surfactant concentration, ultrasonication time, and centrifugation rate. The resulting graphene exhibits exceptional electrical conductivity, enabling its utilization for enhancing the electrical properties of FTO samples.

Consequently, the G-FTO films demonstrate low electrical sheet resistance without compromising optical transparency, even at minimal graphene loading.

2.2 Thin film deposition

The obtained solution is sprayed on a heated substrate using the Nebulizer, which converts the liquid to mist rather than other spray techniques that produced a huge number of small droplets. So, the prepared films are more homogenous and uniform. The prepared samples of FTO and G-FTO are grown at different times of deposition ranging from 5 to 40 minutes. The optimum time of deposition is 20 min at a constant substrate temperature of 500 °C. The spray will be processed layer by layer with a 5 min interval between each one. The substrate is heated incrementally to the requisite temperature, and the deposition process is allowed to cool at a natural rate to the ambient temperature on a heated plate.

2.3 Measurement and characterization procedures

The structural characteristics of the produced films were investigated by employing X-ray diffraction (XRD) (Philips) with Cu-K α radiation (wavelength 1.540 Å). The film's surface morphology was inspected using a scanning electron microscope (SEM, JEOL JSM-6360 LA). The four-point probe procedure is utilized to characterize the electrical resistivity of the deposited samples. Ultraviolet-visible spectroscopy is utilized to quantify the optical transmittance within a wavelength range of 300 to 900 nanometers.

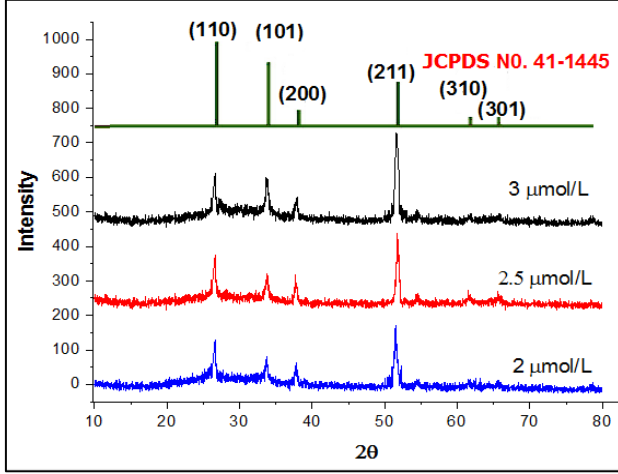
3 Results and discussions

Many parameters can affect the formation of the prepared films of FTO and G-FTO. These parameters are classified into two categories, which are fixed and variable parameters. The fixed parameters were kept constant during the preparation process such as the solution flow rate (1.5 mL/min.), the substrate temperature (500 °C), and the separated distance between the Nebulizer aperture and the sample (5 cm). On the other hand, there are two variables' parameters: concentration of the solution and the time of deposition. For the solution concentration, more than one sample of FTO and G-FTO are prepared at different fluorine concentrations (2, 2.25, 2.5, 2.75, and 3 $\mu\text{mole/L}$) and different graphene concentrations (0.2, 0.3, 0.4, 0.5, and 0.6 $\mu\text{mole/L}$). The addition of fluorine and graphene to the main solution is to enhance the electrical properties of the prepared films.

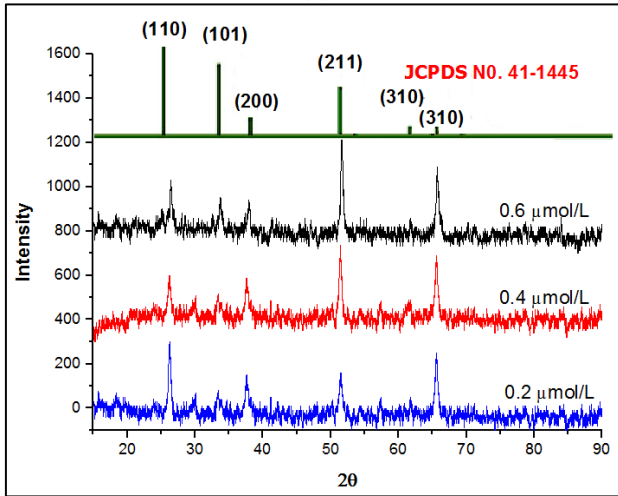
3.1 Structural properties

3.1.1 X-ray diffraction

The crystal structures of FTO and G-FTO thin films are analyzed using X-ray diffraction (XRD). The XRD patterns show different prepared samples of FTO and G-FTO thin films with different fluorine concentrations (2, 2.5 and 3 $\mu\text{mole/L}$) and different graphene concentrations (0.2, 0.4 and 0.6 $\mu\text{mole/L}$).



(a)



(b)

Fig. 1. XRD patterns of (a) FTO and (b) G-FTO doped thin films

The diffraction peaks reveal the polycrystalline nature of the deposited films (Database: JCPDS No. 41-1445). The observed peaks of the FTO and G-FTO thin films are (110), (101), (200), (211), (310), and (301). The FTO and G-FTO thin films exhibit a preferred orientation along the (211) diffraction plane. This dominant orientation peak intensity noticeably increases by increasing the doping level of fluorine and graphene that enhanced the crystallinity of the films as shown in Fig. 1. The lattice constant a and c of the tetragonal phase structures were computed from XRD extracted data using equations [21]

$$2d_{hkl} \sin(\theta) = n\lambda \quad (1)$$

$$\frac{1}{d_{hkl}^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2} \quad (2)$$

where d_{hkl} is the inter-planar distance, (hkl) are the Miller indices, and a and c are the lattice parameters for the tetragonal structure. Their values, listed in Tab. 1, were in good agreement with the standard values of the referred database.

Table 1. Parameters of prepared samples of FTO and G-FTO: inter-planar distance, lattice parameters, grain size, dislocation density, and film thickness extracted for different doping concentrations

CONCENTRATION $\mu\text{mole/L}$	d_{hkl} (\AA)	LATTICE PARAMETERS		GRAIN SIZE D (nm)	$\delta \times 10^4$ (nm^{-2})	THICKNESS (nm)	
		a (\AA)	c (\AA)				
FTO	2	1.76	4.71	3.21	37.2	7.22	238
	2.5	1.77	4.72	3.21	33.4	8.96	780
	3	1.79	4.72	3.22	29.5	11.49	1250
G-FTO	0.2	1.76	4.75	3.24	42.5	5.53	250
	0.4	1.78	4.76	3.25	38.3	6.82	813
	0.6	1.79	4.75	3.25	35.2	8.07	1320

Structure parameters such as grain size D and dislocation density δ have been evaluated also using the XRD data analysis. Deduced values of D and δ , tabulated in Tab. 1, are calculated using Scherrer's formula [22]

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (3)$$

$$\delta = \frac{1}{D^2} \quad (4)$$

Here, λ is the X-ray wavelength 1.5405 \AA , β is referred to the corresponding full-width-at-half maximum (FWHM) of the detected peak, and θ denotes the Bragg angle.

The obtained lattice parameters values *versus* concentration of the doped material of fluorine and graphene are illustrated in Fig. 2. One can observe that there is a contradiction behavior of crystalline size and the doping level. The grain size of these films has been decreased as the concentration of the doped material of fluorine and graphene increases. These reductions may be attributed to the increasing doping level preventing the movement of the grains in the main lattice and limiting the growth of the crystals. It can be observed that D and δ values are increased as the doping concentration increases, as listed in Tab. 1.

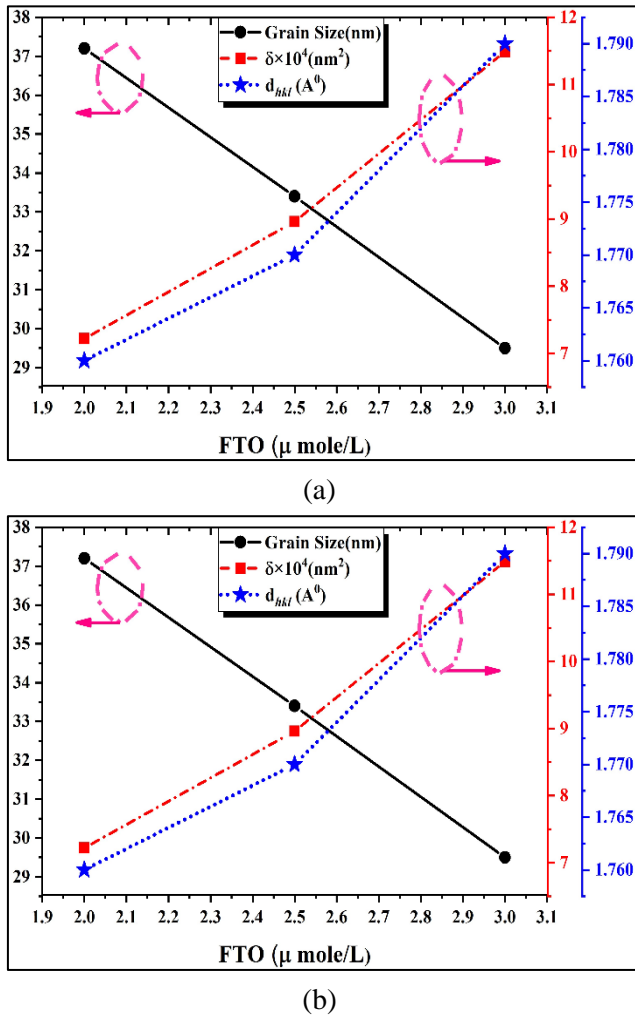


Fig. 2. Doping effect on grain size, dislocation density, and inter-planar distance

3.1.2 Morphological analysis

The surface morphology of FTO and G-FTO thin films deposited with different doping levels of fluorine and graphene concentrations are examined by SEM. The scanning electron microscopy (SEM) images presented in Figures 3 and 4 demonstrate a high degree of homogeneity and uniformity in the grain structure, accompanied by excellent crystallinity. This characteristic is expected to mitigate the adverse effects of grain boundary scattering, resulting in an increase in electrical conductivity.

Notably, the G-FTO thin films exhibit a higher surface roughness and larger grain size compared to the FTO thin films. Analysis using energy-dispersive X-ray spectroscopy (EDX) confirms the presence of distinct

peaks corresponding to the elemental composition of Sn, O, F, and C in both FTO and G-FTO films, indicating that the stoichiometry of the final films remains consistent with FTO and G-FTO compositions.

3.2 Optical properties

The optical transmittance (T) curves of films with different fluorine and graphene concentrations in the wavelength range from 300 to 900 nm are depicted in Fig. 5. The average transmittance of FTO and G-FTO thin films exhibits high optical transmittance around 82% and 80%, respectively. The films demonstrate significant oscillations. This oscillation may be due to the interference phenomenon of the films with the substrate. It is observed that the optical transmission decreases with increasing the fluorine concentration in the solution. This significant reduction may be attributed to the optical scattering and the large grain sizes and boundaries. From the oscillation spectra of optical transmittance curves, the thickness of the films (t) of FTO and G-FTO are determined by using the following relation [23]

$$t = \frac{\lambda_1 \lambda_2}{2[n(\lambda_1)\lambda_2 - n(\lambda_2)\lambda_1]} \quad (5)$$

where $n(\lambda_1)$ and $n(\lambda_2)$ denotes the refractive indices of the adjacent maxima at the two wavelengths λ_1 and λ_2 . In addition, the refractive index (n) can be calculated from the following equations:

$$n = \sqrt{N + \sqrt{(N^2 - n_s^2)}} \quad (6)$$

$$N = \frac{(n_s^2 + 1)}{2} + 2n_s \frac{T_{max} - T_{min}}{T_{max} \cdot T_{min}} \quad (7)$$

where n_s denotes the substrate refractive index. T_{max} and T_{min} denote the maximum and minimum transmittances at the same wavelength as depicted in transmission spectra curves (Fig. 5). The effect of fluorine and graphene concentrations on the thickness of the prepared films is listed in Table 2.

The absorption coefficient (α) can be extracted from the optical transmittance (T) using the following equation [24]

$$T = e^{-\alpha t} \quad (8)$$

where t is the thickness of the FTO and G-FTO films.

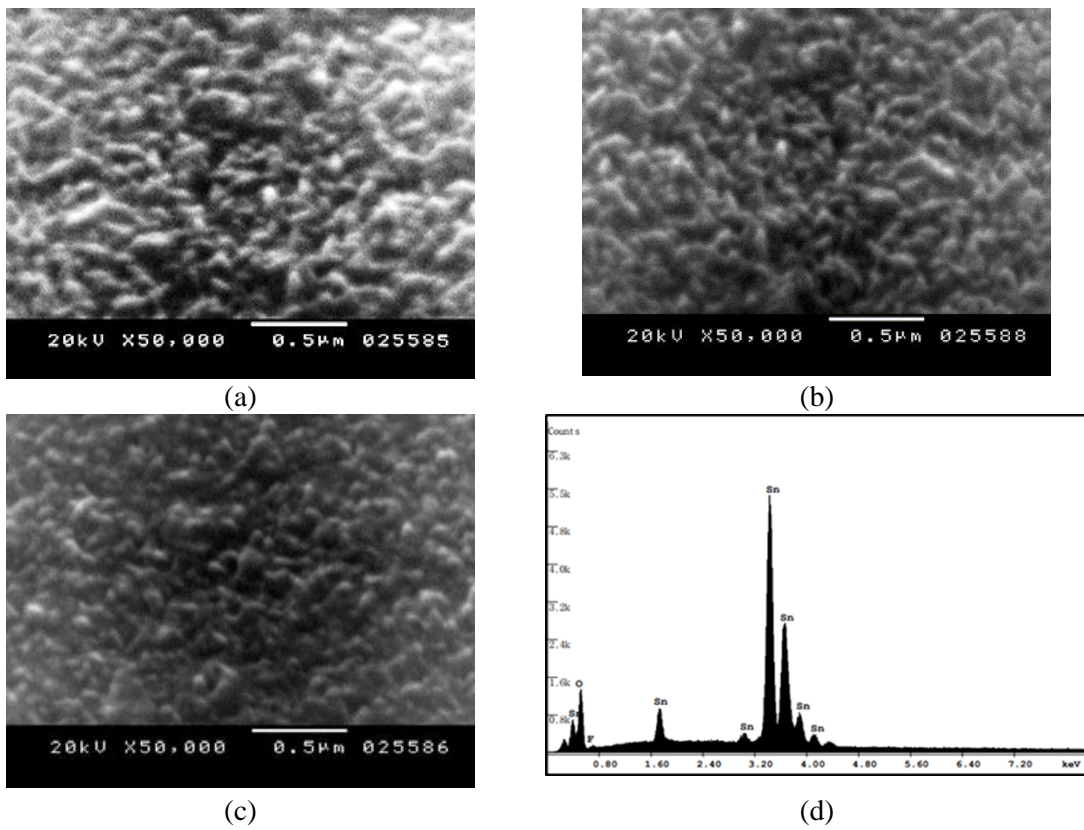


Fig. 3. SEM images and EDX spectrum of FTO: (a) 2 $\mu\text{mole/L}$, (b) 2.5 $\mu\text{mole/L}$, (c) 3 $\mu\text{mole/L}$, and (d) the EDX of FTO film

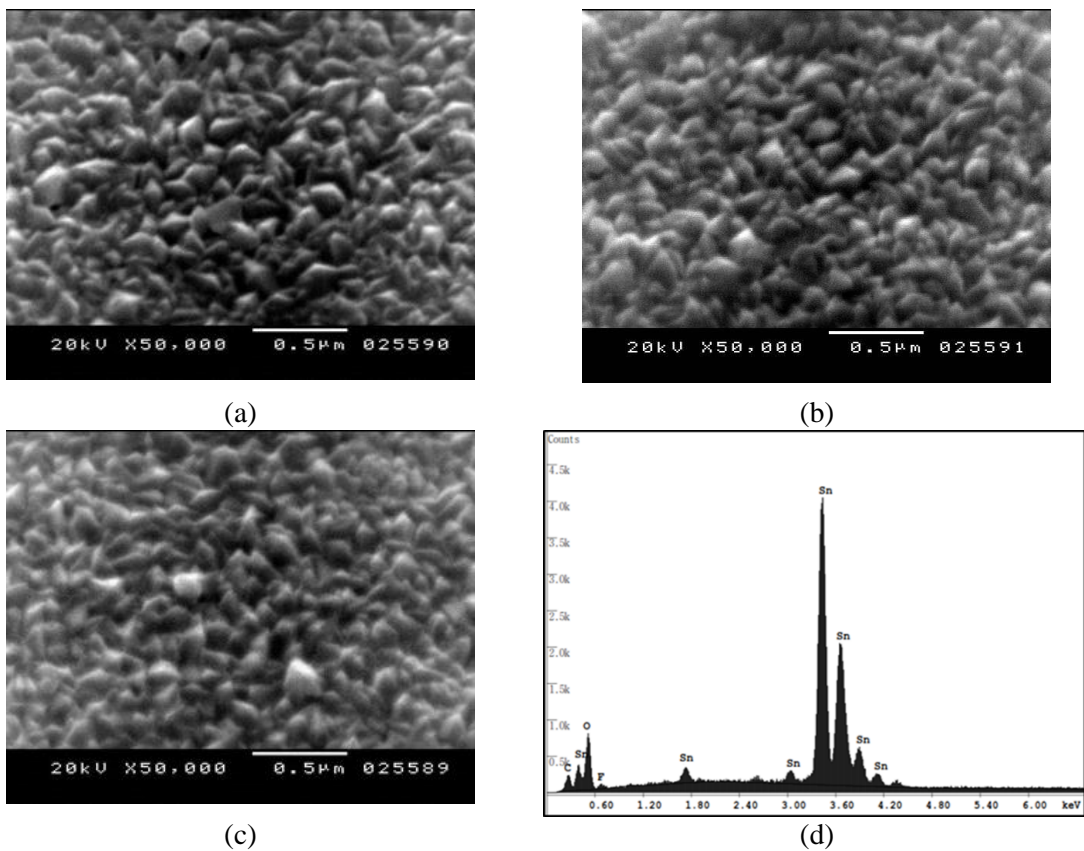
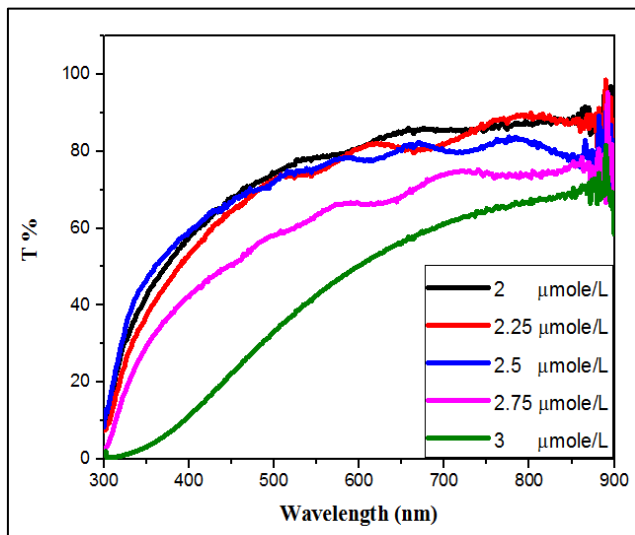


Fig. 4. SEM images and EDX spectrum of G-FTO: (a) 0.2 $\mu\text{mole/L}$, (b) 0.4 $\mu\text{mole/L}$, (c) 0.6 $\mu\text{mole/L}$, and (d) the EDX of G-FTO film

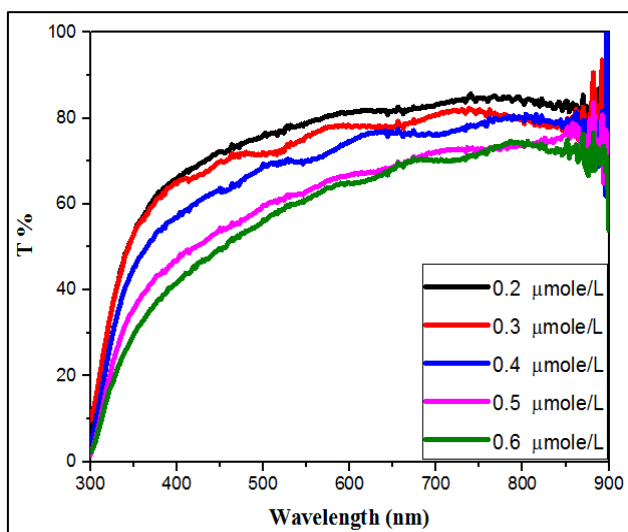
Parameter α is used to determine the optical energy bandgap E_g using the Tauc formula [24]

$$\alpha h\nu = C(h\nu - E_g)^{0.5} \quad (9)$$

where h is the Planck constant, ν is the frequency of the incident photon and C is the constant for direct transition. E_g is estimated by drawing $(\alpha h\nu)^2$ versus $(h\nu)$ and extrapolating the linear portion to the $h\nu$ axis at the point where $(\alpha h\nu)^2 = 0$ [25]. An increase in F doping level in SnO₂ resulted in an increased bandgap, which is consistent with previous studies. This increase is attributed to the Burstein-Moss effect, which is related to an increase in carrier concentration leading to a shift in the conduction band filling [26]. However, E_g values at high concentrations of graphene are reduced due to the low energy gap of the graphene. The energy band gap of the prepared sample of FTO and G-FTO thin films is illustrated in Fig. 6 with different fluorine and graphene concentrations.



(a)



(b)

Fig. 5. Transmission spectra at different fluorine and graphene concentrations of (a) FTO and (b) G-FTO

3.3 Electrical properties

The electrical properties are the dominant parameters that attract our attention in this work. The addition of fluorine to pure SnO₂ enhances the electrical conductivity of the prepared FTO thin films. Therefore, the variations in fluorine concentration are controlling the value of sheet resistance R_{sh} measured by a four-point probe. In this procedure, the electric current I is injected through the outer two probes while the drop voltage V is measured between the inners, thus an accurate estimation of sheet resistance R_{sh} obtained by [27]

$$R_{sh} = 4.532 \left(\frac{V}{I} \right) \quad (10)$$

Firstly, the sheet resistance decreases as the fluorine concentration increases until attaining a minimum value then the sheet resistance raises up again. At 2.5 μmole/L of fluorine, the lowest sheet resistance is 15 Ω/□ and increases up to 120 Ω/□ as the fluorine concentration increases up to 3 μmole/L and this result is in agreement with that obtained from optical properties [28]. In FTO thin films, F atoms substitute the O atoms in the main lattice creating extra free electrons and therefore enhancing the electrical properties of the prepared samples. The overabundant and excess of F atoms in the precursor solution increases the free carrier concentration. The motion of these carriers decreases by increasing the scattering between them leading to the increase of the sheet resistance [29].

The figure of merit (FOM) is an appropriate quantitative measure that can be used to rate the performance of the evaluated object. Measuring that can rate the performance of the evaluated object and determine its comparative effectiveness for an application. Several figures of merit have so far been defined by researchers for rating the performance of TCO which can be used for solar cells and optoelectronic applications. FOM (ϕ_m) is a vital parameter in evaluating FTO thin films. It is calculated by using the Haacke formula [30]

$$\phi_m = \frac{T^{10}}{R_{sh}} \quad (11)$$

where T is the optical transmittance and R_{sh} is the electrical sheet resistance. According to Eqn. (11), this FOM becomes large by maximizing T and minimizing R_{sh} as shown in Table 2. The optimum preparation conditions of FTO thin films are determined using the FOM that produces high optical transmittance and low electrical sheet resistance. The optimum FTO thin film is obtained from a sample of 2.5 μmole/L fluorine concentration at 500°C of substrate temperature and 20 minutes of spraying time to give an R_{sh} value of 15 Ω/□ and T value of 82% with FOM of $91.2 \times 10^{-4} \Omega^{-1}$. For the G-FTO thin films; different graphene concentrations (0.2, 0.3, 0.4, 0.5, and 0.6 μmole/L) are added to the precursor 2.5 μmole/L of the optimum FTO sample. The

addition of graphene to the main solution exhibited a significant impact on enhancing the properties of the prepared samples. The optical transmittance curves of G-FTO thin films are illustrated in Fig. 5 revealing that the addition of graphene to the FTO sample does not affect critically the optical properties that range from 70 to 85 % according to the graphene concentrations but the electrical properties of FTO thin films is the main parameter that affected by the addition of graphene to the main solution. The sheet resistance of the G-FTO thin films is reduced compared to that of the FTO as displayed in Table 2. The FOM of the prepared samples of G-FTO reveals that the optimum concentration of the graphene material that gives the best behavior (higher transmittance, lower electrical sheet resistance) is at 0.4 $\mu\text{mole/L}$ of graphene concentration, 500°C of substrate temperature, and 20 min of spraying time and give an R_{sh} value of 8 Ω/\square , transmittance of 79%, and FOM of $118.3 \times 10^{-4} \Omega^{-1}$.

It is important to mention that R_{sh} is related to the electrical conductivity (σ) and the thin film thickness using the relation [31]:

$$\sigma = q \cdot n \cdot \mu \tag{12}$$

$$R_{sh} = \frac{1}{\sigma t} \tag{13}$$

where n is the free electron concentration, q is the charge of the electron and μ is the electron mobility. As the electron mobility is governed by the grain boundary scattering that is represented by [31]

$$\mu = \frac{D \cdot q}{(2\pi m_n^* k_B T)^{1/2}} \exp\left(\frac{-\varphi_b}{k_B T}\right) \tag{14}$$

where D is the grain size, m_n^* is the effective mass of electrons, φ_b is the grain boundary potential barrier, k_B is the Boltzmann constant, and T is the temperature. In the case of the grain boundary scattering, the sheet resistance can be represented by

$$R_{sh} = \frac{1}{q \cdot n \cdot \mu \cdot t} = \frac{1}{M \cdot D} \tag{15}$$

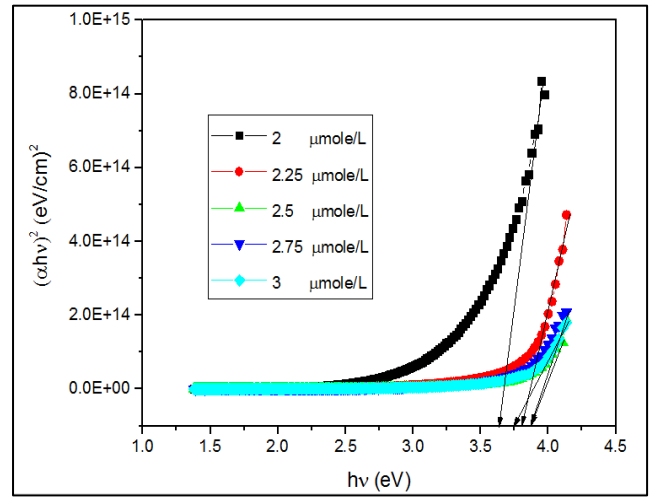
where

$$M = \frac{n \cdot q^2 \cdot t}{(2\pi m_n^* k_B T)^{1/2}} \exp\left(\frac{-\varphi_b}{k_B T}\right). \tag{16}$$

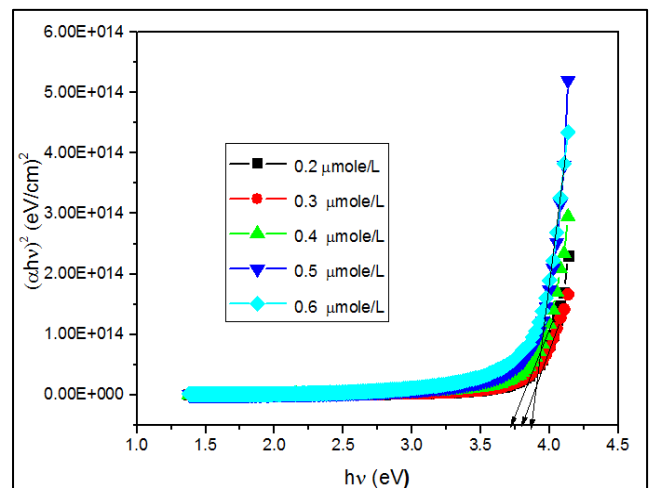
Equation (15) indicates an inverse relationship between the sheet resistance and the grain size of the samples. This trend agrees with results of the XRD analysis which gives a higher grain size in the G-FTO films of 38.3 nm with a lower sheet resistance of 8 Ω/\square . In comparison of the FTO films give a grain size of 33.5 nm with a sheet resistance of 15 Ω/\square .

This result coincides with the SEM measurements that reveal the good homogeneity and crystallinity of G-FTO samples rather than the FTO as indicated in Figs. 3 and 4.

The present study indicates that the solvent concentration can greatly affect the physical properties of FTO and G-FTO thin films. So, highly transparent conducting oxide thin films of FTO and G-FTO with low electrical sheet resistance were fabricated via the nebulizer spray pyrolysis technique. In this regard, our preparation method and proposed doping procedure introduce the material as a good candidate for optoelectronic applications. For the sake of showing the effectiveness of the proposed thin films, a comparison between recently published studies and our prepared samples of FTO and G-FTO is listed in Table 3. It is observed that the addition of graphene to the main solution enhanced the overall electrical properties while the optical transmission is not significantly affected.



(a)



(b)

Fig. 6. Optical energy band gap at different doping levels of (a) FTO and (b) G-FTO

Table 2. Influence of doping concentration on thickness, transmittance, electrical and optical properties of FTO and G-FTO thin films

CONCENTRATION ($\mu\text{mole/L}$)	THICKNESS (nm)	T (%)	R_{sh} (Ω/\square)	RESISTIVITY $\rho \times 10^{-6}$ ($\Omega \cdot \text{cm}$)	E_g (eV)	$\phi_m \times 10^{-4}$ (Ω^{-1})	
FTO	2	238	89	130	30.1	3.6	23.9
	2.25	450	85	45	20.25	3.7	43.7
	2.5	780	82	15	11.7	3.8	91.6
	2.75	966	74	50	48.3	3.9	9.8
	3	1250	66	120	150	3.7	1.3
G-FTO	0.2	250	85	45	11.2	3.7	43.7
	0.3	620	82	25	15.5	3.8	54.9
	0.4	813	79	8	6.5	3.9	118.3
	0.5	990	72	30	29.7	3.8	12.4
	0.6	1320	70	47	62.1	3.7	6.01

Table 3. Comparison of results obtained in this work to those reported in recently published reports

REF.		THICKNESS (nm)	T (%)	R_{sh}	E_g (eV)	$\phi_m \times 10^{-4}$ (Ω^{-1})
[32]	G-FTO	-	77.3	26	3.69	28.3
[33]	FTO	475	81.2	24	4.01	51.9
[27]	FTO	700	81	21	3.84	57.9
[34]	FTO	320	88.5	40	3.96	73.6
[35]	FTO	-	68.5	4.5	-	50.5
[36]	FTO	620	78	-	3.75	11.5
[37]	FTO	350	85	18.9	-	130
[38]	G-FTO	300	75	372	-	15.1
Present work	FTO	780	82	15	3.8	91.6
	G-FTO	813	79	8	3.9	118.3

4 Conclusion

Special kinds of transparent conducting oxide thin films of fluorine-doped tin oxide (FTO) and graphene-fluorine-doped tin oxide (G-FTO) were successfully deposited with different fluorine and graphene concentrations at a substrate temperature of 500 °C using low-cost spray pyrolysis deposition technique. The XRD analysis exhibited that the prepared FTO and G-FTO thin films had a tetragonal structure and were polycrystalline. Scanning electron microscope images demonstrated high homogenous grains, smooth surface, and good uniformity of the deposited sample of G-FTO rather than the FTO. The measured EDX spectrum of the films showed the presence of Sn, O, C, and F elements. The optimum FTO film was obtained from 2.5 $\mu\text{mole/L}$ of fluorine concentrations with 82% optical transmittance and 15 Ω/\square sheet resistance with FOM of $91.6 \times 10^{-4} \Omega^{-1}$. In addition, the optimum G-FTO thin film with 79% optical transmittance, 8 Ω/\square of electrical sheet resistance, and FOM of $118.3 \times 10^{-4} \Omega^{-1}$ was fabricated from 0.4 $\mu\text{mole/L}$ of graphene concentration. The energy bandgap due to its direct transition varied from 3.6 to 3.9 eV in the prepared

samples. According to the structural, morphological, optical, and electrical properties, the prepared G-FTO film is a good candidate for optoelectronic applications.

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Received 12 August 2023